

# Synthesis and microwave dielectric properties of $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$ phases

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## Abstract

Oxides of the type,  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$  ( $0 \leq x \leq 1$ ) have been obtained by the ceramic method. These oxides crystallize in the hexagonal cell corresponding to ordered triple perovskites. Sintered disks show nearly frequency-independent dielectric constant for all the compositions. Compositions sintered at 1425°C yield dielectric constant of 20–22 at ~6 GHz, with quality factor ranging from 1300 to 1500.  $\text{Sr}_3\text{Zn}_{0.5}\text{Mg}_{0.5}\text{Nb}_2\text{O}_9$  shows a very low temperature coefficient of resonant frequency ( $\tau_f$ ) of +4 ppm/°C.

*Keywords:* A. Ceramics; A. Electronic materials; A. Oxides; C. X-ray diffraction; D. Dielectric properties

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## 1. Introduction

In perovskite compounds of the formula  $\text{A}(\text{B}_{1/3}\text{B}'_{2/3})\text{O}_3$  an ordered phase with 1:2 regular arrangement of the B and B' ions often exists. The structure of the ordered phase has been assigned by Galasso and Pyle [1]. Some of the compounds with an ordered configuration of B and B' ions on the octahedral sites having a low dielectric loss are used as dielectric resonators. The dielectric resonator has the advantages of compactness and ease of matching to microwave-integrated circuits. The desired

ceramic characteristics for use as microwave resonators include a suitable dielectric constant  $\epsilon$  (20–40), high unloaded  $Q$  (>3000) and low-temperature coefficient of the resonant frequency ( $\tau_f$ ) 0 + 10 ppm/ $^\circ$ C.

In the past 15 years, a number of ceramics have been developed [2–4] as microwave dielectric resonators like  $\text{BaZn}_{1/3}\text{Nb}_{2/3}\text{O}_3$ ,  $\text{BaMg}_{1/3}\text{Nb}_{2/3}\text{O}_3$  and  $\text{BaZn}_{1/3}\text{Ta}_{2/3}\text{O}_3$ . Substitution of the Ba-site with Sr has been attempted earlier to form solid solutions and their dielectric properties have also been investigated. For example, the system  $\text{Ba}(\text{Mg}_{1/3}\text{Ta}_{2/3})\text{O}_3$ – $\text{Sr}(\text{Mg}_{1/3}\text{Ta}_{2/3})\text{O}_3$  [5] and  $\text{Ba}(\text{Zn}_{1/3}\text{Nb}_{2/3})\text{O}_3$ – $\text{Sr}(\text{Zn}_{1/3}\text{Nb}_{2/3})\text{O}_3$  [6], have been systematically explored to obtain new materials with better dielectric properties.  $\text{Sr}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3$  was reported to be hexagonally ordered around 1650 $^\circ$ C [7].

In this paper, we describe the structural and dielectric properties of solid solutions of the type  $\text{Sr}(\text{Zn}_{1-x}\text{Mg}_x)_{1/3}\text{Nb}_{2/3}\text{O}_3$ , where the effect of B-site substitution has been investigated in the Sr-based perovskite oxide.

## 2. Experimental

The starting materials were  $\text{SrCO}_3$ ,  $\text{MgO}$ ,  $\text{ZnO}$  and  $\text{Nb}_2\text{O}_5$  powders. They were mixed and milled with a mortar and pestle. The mixed powders were then calcined at 1000 $^\circ$ C for 30 h with intermittent grinding. The calcined powders were mixed with 5% of polyvinylalcohol (PVA) and made into disks. The disks were then sintered at 1000 $^\circ$ C for 48 h, 1200 $^\circ$ C for 12 h and finally at 1300 $^\circ$ C for 12 h. For measurement at GHz frequencies the samples were further sintered at 1425 $^\circ$ C for 6 h. The powders were studied by powder X-ray diffraction (XRD) (Bruker D8—Advance diffractometer). Particle size of the sintered materials were measured using a Scanning Electron Microscope (Cambridge 360). The density of the sintered disks was measured by Archimedes method. The as-sintered samples were coated with “Al” in vacuum to measure the dielectric constant and loss tangent in the low frequency range (50 Hz–500 kHz) using a LCR meter (HP 4284L). Dielectric properties at GHz frequencies were measured using a Network Analyzer (Hewlett-Packard 8510B) and a reflection transmission unit (Hewlett-Packard 8510B). The apparatus was controlled using a Hewlett-Packard 9000, 300 series computer. The  $\epsilon_r$  was calculated from the  $\text{TE}_{011}$  resonance mode of the end-shortened sample placed between two conducting plates, using the method of Hakki and Coleman [8] and modified by Courtney [9]. The  $Q$ -factor was measured by the microstripline method of Khanna and Garault [10]. The coefficient of temperature variation of resonant frequency was measured by noting the temperature variation of resonant frequency of the  $\text{TE}_{011}$  mode in the transmission configuration over a range of temperature 23–80 $^\circ$ C.

## 3. Results and discussion

The entire range of compounds in the  $\text{Sr}_3\text{ZnNb}_2\text{O}_9$ – $\text{Sr}_3\text{MgNb}_2\text{O}_9$  system crystallize in a cubic structure when calcined at 1000 $^\circ$ C. The lattice parameter ( $a$ ) of these

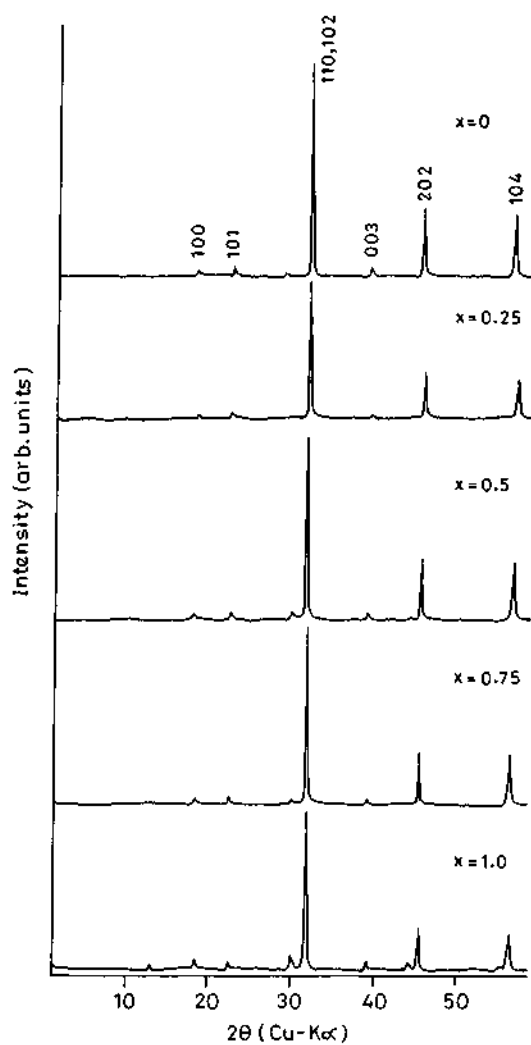


Fig. 1. Powder X-ray diffraction patterns of  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$  system sintered at  $1300^\circ\text{C}$ .

Table 1

Lattice parameters ( $\text{\AA}$ ) for the oxides of the type  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$

Composition	$a_{\text{cubic}}^{\text{a}}$	Ordered phase (hexagonal) <sup>b</sup>	
		$a$	$c$
$\text{Sr}_3\text{ZnNb}_2\text{O}_9$	4.0067(2)	5.646(6)	6.907(3)
$\text{Sr}_3\text{Zn}_{0.75}\text{Mg}_{0.25}\text{Nb}_2\text{O}_9$	4.0058(3)	5.653(5)	6.913(3)
$\text{Sr}_3\text{Zn}_{0.5}\text{Mg}_{0.5}\text{Nb}_2\text{O}_9$	4.0059(3)	5.653(5)	6.918(3)
$\text{Sr}_3\text{Zn}_{0.25}\text{Mg}_{0.75}\text{Nb}_2\text{O}_9$	4.0048(1)	5.653(3)	6.919(2)
$\text{Sr}_3\text{MgNb}_2\text{O}_9$	4.0006(4)	5.643(4)	6.909(2)

<sup>a</sup>  $1000^\circ\text{C}/30$  h.

<sup>b</sup>  $1000^\circ\text{C}/30$  h/ $1000^\circ\text{C}/48$  h/ $1200^\circ\text{C}/24$  h/ $1300^\circ\text{C}/12$  h.

compounds is  $\sim 4.001 \text{ \AA}$ . Powder XRD of the  $1300^\circ\text{C}$  sintered samples show the hexagonally ordered phase (Fig. 1) for all the oxides. All the reflections could be indexed satisfactorily in the hexagonal cell with  $a \sim 5.65 \text{ \AA}$  and  $c \sim 6.91 \text{ \AA}$ . However, the XRD patterns show a small amount of ( $\sim 5\%$ )  $\text{Sr}_5\text{Nb}_4\text{O}_{15}$  phase as

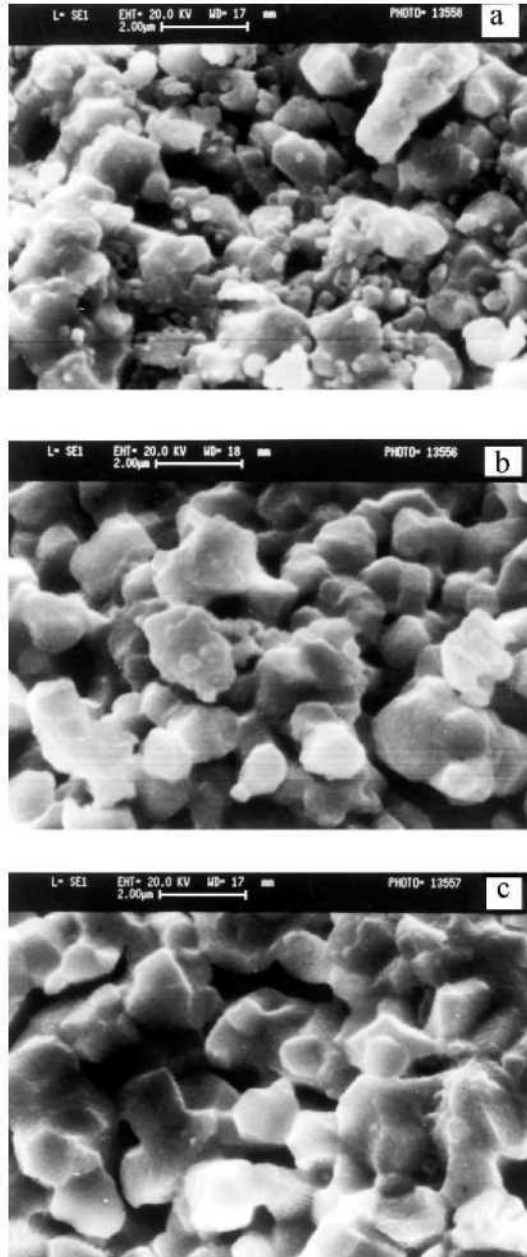


Fig. 2. Scanning electron micrographs of oxides of the  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$  system sintered at  $1425^\circ\text{C}$  for (a)  $x = 0.25$ ; (b)  $x = 0.50$ ; (c)  $x = 0.75$ .

impurity. The lattice parameters and other X-ray data are summarized in Table 1.  $\text{Sr}_3\text{ZnNb}_2\text{O}_9$  is earlier reported to be hexagonally ordered with  $a = 5.66 \text{ \AA}$ ,  $c = 6.95 \text{ \AA}$  [6]. These values are slightly higher than what we obtained (Table 1). There is a very small decrease in the lattice parameters as we substitute Mg in Zn sites

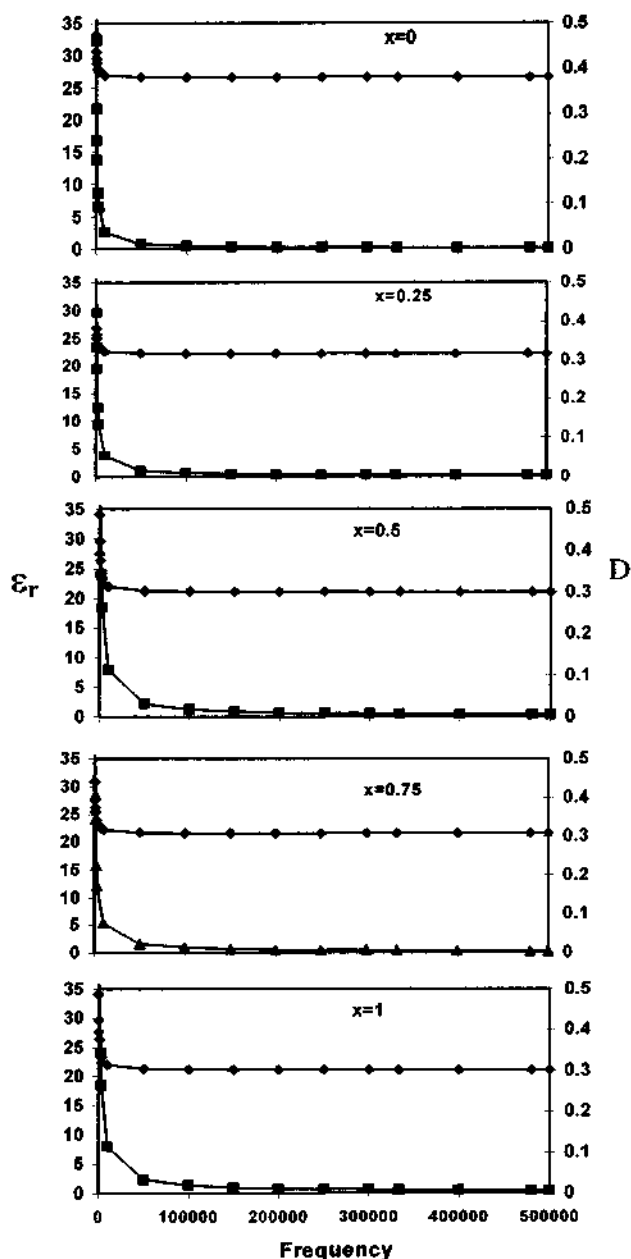


Fig. 3. Variation of dielectric constant ( $\epsilon_r$ ) and dielectric loss ( $D$ ) with frequency for  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$  system in the 50 Hz–500 kHz region.

Table 2  
Dielectric properties for  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$  at GHz frequencies

Composition	$\epsilon_r$	$Q \times f$ ( $f$ in GHz)	$\tau_f$ (ppm/ $^\circ\text{C}$ )
$\text{Sr}_3\text{ZnNb}_2\text{O}_9$	20.69	$1300 \times 5.8197 = 7565$	-27.43
$\text{Sr}_3\text{Zn}_{0.75}\text{Mg}_{0.25}\text{Nb}_2\text{O}_9$	19.96	$1429 \times 5.9750 = 8538$	-15.7
$\text{Sr}_3\text{Zn}_{0.5}\text{Mg}_{0.5}\text{Nb}_2\text{O}_9$	22.31	$1521 \times 5.3665 = 8162$	+4.38
$\text{Sr}_3\text{Zn}_{0.25}\text{Mg}_{0.75}\text{Nb}_2\text{O}_9$	21.55	$1168 \times 5.7525 = 6718$	-10.23

$\epsilon_r$ : dielectric constant;  $Q$ : quality factor;  $\tau_f$ : temperature coefficient of resonant frequency.

and is observable at high “Mg” concentrations. This is understood easily since the ionic radii [11] of  $\text{Mg}^{2+}(\text{VI})$  is 0.720 and that of  $\text{Zn}^{2+}(\text{VI})$  is 0.740.

The scanning electron micrograph of the samples sintered at  $1425^\circ\text{C}$  are shown in Fig. 2. The average grain size (calculated by the linear intercept method) is around  $1 \mu\text{m}$  for  $x = 0.25$ . We see a distribution of grain sizes. However, for  $x = 0.5$  and  $0.75$ , the grains are more regular and compact. The average grain size also increases to  $1.8$  and  $2.2 \mu\text{m}$  for  $x = 0.5$  and  $0.75$  compositions, respectively.

The room temperature dielectric properties on sintered pellets ( $1300^\circ\text{C}$ ) show frequency stable (Fig. 3) dielectric constant around 20–27 for all the compositions in the 100 Hz–500 kHz range. The dielectric loss was  $\sim 0.001$  for all compositions. Both the dielectric constant and the dielectric loss remained almost constant in the entire frequency range measured. There have been earlier reports [6] of dielectric properties of SZN and SMN, where the dielectric constant is reported to be 37–40, respectively at 100 kHz. However, no detail of the X-ray studies are given to ensure absence of impurities. In addition, the variation of the dielectric constant with frequency has also not been studied earlier [6].

We have also studied the dielectric properties at microwave frequencies (5.4–5.9 GHz) for disks sintered at  $1425^\circ\text{C}$ . The observed dielectric constant and quality factor for various members of the solid solution are shown in Table 2. The density of these samples were  $\sim 97\%$  of the theoretical density. The dielectric constant varies between 20 and 22 for various members of the  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$  family. The  $x = 0.5$  composition,  $\text{Sr}_3(\text{Zn}_{0.5}\text{Mg}_{0.5})\text{Nb}_2\text{O}_9$ , has the highest dielectric constant of 22.3. All the samples have low loss or a high  $Q$ . The  $Q \times f$  values are listed in Table 2. The temperature coefficient of resonant frequency ( $\tau_f$ ) is quite low, varying between  $-27$  and  $+4.3 \text{ ppm}/^\circ\text{C}$  for different compositions. The  $x = 0.5$  composition has the best dielectric properties among the compositions studied with the highest  $\epsilon_r$ , lowest loss and lowest  $\tau_f$  value of  $+4.38 \text{ ppm}/^\circ\text{C}$ . Thus, our studies show the possibility of fine tuning the properties by suitable substitution in the course of which materials with nearly zero  $\tau_f$  can be obtained.

#### 4. Conclusions

The detailed structural studies using powder XRD and the dielectric properties at low frequency (50 Hz–500 kHz) and at high frequency (5–6 GHz) was investigated

for the  $\text{Sr}_3\text{Zn}_{1-x}\text{Mg}_x\text{Nb}_2\text{O}_9$  system. All the oxides are hexagonally ordered after sintering at  $1300^\circ\text{C}$ . These oxides have extremely stable dielectric constant in the 100 Hz–500 kHz frequency range. From microwave (5.4 GHz) measurement it is found that the  $x = 0.5$  composition has a very small temperature coefficient of the resonant frequency ( $\tau_f$ ) of  $+4.3 \text{ ppm}/^\circ\text{C}$ . It has a relatively high dielectric constant of 22.3 and a  $Q$  of 1521. This composition has characteristics required for use as a dielectric resonator at microwave frequencies.

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